



# Facile fabrication of heterostructured BiPS<sub>4</sub>-Bi<sub>2</sub>S<sub>3</sub>-BiVO<sub>4</sub> photoanode for enhanced stability and photoelectrochemical water splitting performance

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## ARTICLE INFO

### Article history:

Received 22 May 2022

Revised 14 December 2022

Accepted 27 December 2022

Available online 30 December 2022

### Keywords:

Bismuth vanadate

Photo-electro-transformation

Bismuth Sulfide

Metal Phosphorous Trichalcogenides

Electrolyte tuning

## ABSTRACT

Bismuth vanadate (BiVO<sub>4</sub>) is the most favorable electrode candidate for photoelectrochemical (PEC) water-splitting reactions. The poor charge separation and sluggish water oxidation dynamics are, however, the major setbacks of BiVO<sub>4</sub> photoanodes. To address these issues, we demonstrate that bismuth thiophosphate (BiPS<sub>4</sub>)-Bi<sub>2</sub>S<sub>3</sub> hybrid nanostructure was photoelectrochemically transformed on BiVO<sub>4</sub> electrodes (BiPS<sub>4</sub>-Bi<sub>2</sub>S<sub>3</sub>-Bi<sub>2</sub>O<sub>3</sub>) when treated in Na<sub>2</sub>S/PBS electrolyte, and a notable photocurrent of 3.5 mA/cm<sup>2</sup> at 0.65 V<sub>RHE</sub> was obtained showing promising stability. X-ray photoelectron spectroscopy (XPS) and scanning electron microscopy (SEM) examination evidenced the effective makeover of BiVO<sub>4</sub> into the BiPS<sub>4</sub>-Bi<sub>2</sub>S<sub>3</sub>/BiVO<sub>4</sub> nanostructured matrix. A negatively shifted onset potential and enriched durability are achieved for heterostructured BiPS<sub>4</sub>-Bi<sub>2</sub>S<sub>3</sub>/BiVO<sub>4</sub> photoanodes due to decreased surface recombination. Interestingly, the Bode phase analysis evidenced the faster hole consumption in the water oxidation process in the BiPS<sub>4</sub>-Bi<sub>2</sub>S<sub>3</sub>/BiVO<sub>4</sub> electrode compared to pristine BiVO<sub>4</sub>. This methodology can be engaged to design different complex nanostructured materials with tunable optical and electrical features for photoelectrocatalysis, electrical energy storage, and solar cell uses.

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## 1. Introduction

Solar radiation is a virtually inexhaustible and clean energy resource. Solar H<sub>2</sub> generation is accepted as a highly favorable technology to exploit solar energy and produce sustainable chemical fuels, tackling climate change stemming from the combustion of fossil fuels [1]. However, the efficient future use of solar energy to satisfy global energy demand is still dependent on developing efficient devices for energy conversion, storage, and distribution. The transformation of solar energy into hydrogen fuel, a clean and storable fuel, is a promising and active technological research field. Solar-driven water photo electro-catalysis (PEC) using semi-

conductor materials to produce hydrogen fuel has been a leading approach [2]. It is compulsory to develop highly efficient electrode materials that absorb a major portion of the solar irradiance, with suitable conduction band (CB) edge levels, chemical inertness, and cost-effectiveness.

Up to now, various *n*-type semiconductor photoanodes have been advanced for solar hydrogen production, such as TiO<sub>2</sub> [3], Fe<sub>2</sub>O<sub>3</sub> [4], WO<sub>3</sub> [5], and BiVO<sub>4</sub> [6]. In particular, *n*-type BiVO<sub>4</sub> is one of the most attractive photoanodes owing to its favorable optoelectronic and photoelectrochemical properties such as relatively narrow bandgap, natural inertness, and appropriate energy level positions for water oxidation, and cost-effective [7–8]. The maximum theoretical photocurrent for BiVO<sub>4</sub> electrodes, with its narrow bandgap of ~2.4 eV, is ~7.5 mA/cm<sup>2</sup> at 1.23 V<sub>RHE</sub>. Also, BiVO<sub>4</sub> can obtain a theoretic Solar-To-Hydrogen efficiency of 9.2 % [9]. Though, the most severe issue associated with BiVO<sub>4</sub> catalysts is the recombination of charge carriers, limiting its photocat-

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alytic efficiency as it results in a relatively short carrier diffusion length (ca. 70 nm) [10–11]. In literature, the identified photocurrent response for the bare BiVO<sub>4</sub> electrodes is thus generally < 1 mA/cm<sup>2</sup> at 1.23 V<sub>RHE</sub> [12]. To overcome these concerns, different approaches such as facet modifications [13], morphology control [14], impurity doping [15–17], oxygen evolution catalyst deposition [18–19], and construction of p-n or n-n heterostructures [20] have been developed to upsurge the photon absorption and carrier transport of BiVO<sub>4</sub>. More importantly, developing type II heterostructures is a favorable route to boost the charge transfer efficiency by creating an effective charge cascade and improving the visible-light absorption with other materials. For instance, Ying et al. fabricated BiVO<sub>4</sub>/Bi<sub>2</sub>O<sub>3</sub> heterostructured photoanodes by solvothermal synthesis and examined the photodegradation of methyl orange under illumination [21]. Further, Chae group reported WO<sub>3</sub>/BiVO<sub>4</sub> photoanodes to boost the efficiency of solar-driven water splitting by improving the charge transfer and light absorption ability [22]. Regmi et al. demonstrated that hydrothermally fabricated BiOBr/BiVO<sub>4</sub> photoanodes improve malachite green degradation efficiency under visible-light excitation [23]. More recently, Ta et al. demonstrated the 3D-structured WO<sub>3</sub>/Mo:BiVO<sub>4</sub> heterostructured photoanodes for PEC water oxidation reaction under acidic and neutral aqueous electrolytes. Moreover, we have developed BiVO<sub>4</sub>/AgPi [24], BiVO<sub>4</sub>/Ni-Fe Prussian blue [25], BiVO<sub>4</sub>/Ag-Ni-OH-Pi [26], and SnO<sub>2</sub>/BiVO<sub>4</sub>/NiWO<sub>4</sub> [27] heterojunction-based photoanodes, showing enhanced performance compared to bare BiVO<sub>4</sub>, owed to the well-separated and long-life photoinduced charge carriers for redox reaction offered by the developed heterojunctions. Since the oxygen Evolution Reaction (OER) is the most critical and so determining step toward the generation of solar Hydrogen from water splitting, the deposition of suitable co-catalytic materials in photocatalytic OER is mandatory. In this context, several research works combine OER catalysts over BiVO<sub>4</sub> photoanodes to improve charge transfer kinetics. For instance, CoPi, CoO<sub>x</sub>, and NiFeO<sub>x</sub> electrocatalysts are decorated onto BiVO<sub>4</sub> electrodes [28–31], decreasing the surface recombination at the interface and/or enhancing OER kinetics.

On the other hand, Bi<sub>2</sub>S<sub>3</sub> with a low bandgap of 1.4 eV has an appropriate energy level position that matches well with BiVO<sub>4</sub> for efficient charge separation and can be coupled with BiVO<sub>4</sub> to develop heterostructured photoanodes [32]. Indeed, there are some seminal studies of BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> heterostructured electrodes for PEC water splitting. Notably, Liu et al. demonstrated BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> nanowires heterostructured photoanode via a hydrothermal route and achieved a photocurrent response of 0.51 mA/cm<sup>2</sup> [33]. Zhuo et al. showed walnut-like BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> heterostructured photoanodes with improved photocatalytic performance [34]. Particularly, Chen et al. designed BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> heterostructured electrodes over conducting films and reached a remarkable photocurrent response of 1.43 mA/cm<sup>2</sup> under illumination [35]. On another hand, BiPS<sub>4</sub> is a solar absorber with suitable energy level positions that also match BiVO<sub>4</sub> for efficient electron-hole pairs separation [36]. These studies inspired us to explore BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> triple-layer heterostructured photoanodes as promising electrodes for PEC water splitting.

Herein, we describe a controlled electrochemical ion exchange strategy for producing a thin and homogenous Bi<sub>2</sub>S<sub>3</sub>/BiPS<sub>4</sub> film transformed over the surface of BiVO<sub>4</sub> electrodes, and its PEC features for water oxidation. The synthesized BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> heterostructured electrodes demonstrated a photocurrent response of ~ 3.7 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub> showing almost 5-fold enhancement related to bare BiVO<sub>4</sub> electrodes. The development of heterostructured electrodes based on BiVO<sub>4</sub> and Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> has stronger light absorption, enhanced carrier separation, and remarkable transport and surface features.

## 2. Materials and methods

### 2.1. Materials

Bismuth(III) nitrate [Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O, ≥99.0 %], vanadyl acetylacetonate [VO(acac)<sub>2</sub>, ≥98.0 %], Sodium sulfide (Na<sub>2</sub>S, ≥99.0 %), sodium sulfite (Na<sub>2</sub>SO<sub>3</sub>, ≥99.0 %), sodium sulfate (Na<sub>2</sub>SO<sub>4</sub>, ≥99.0 %), Dimethylsulfoxide (DMSO, ≥ 99.9 %) and ethylene glycol (EG, ≥99 %) were received from Sigma Aldrich. A Millipore system was adapted to acquire high-quality water to prepare the electrolyte solutions.

### 2.2. Fabrication of BiVO<sub>4</sub> photoanodes

BiVO<sub>4</sub> photoelectrodes were obtained via dual deposition process described by Choi et al [37]. An EG solution comprising 20 mM of Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O was acquired as the plating solution. The electrodeposition was executed by applying 0.04C/cm<sup>2</sup> at E = -1.8 V vs Ag/AgCl, followed by a relaxing time of 2 s. Subsequently, this electrochemical cycle was continual 8 times to give an entire charge of 0.40C/cm<sup>2</sup>. Then, 60 μL of a DMSO solution involving 0.2 M VO(acac)<sub>2</sub> was kept over the as-deposited Bi electrode to completely protect their surface and was then subjected to a muffle furnace at 500 °C for 2 h in the air (rate = 2 °C/min) to convert Bi to BiVO<sub>4</sub>. Left-over VO<sup>+2</sup> ion was eliminated by drenching them in 1 M NaOH for 20 min and dried in air.

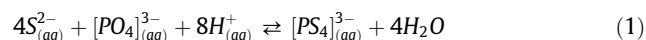
### 2.3. Fabrication of BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> heterostructured photoanodes

The BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> heterostructured photoanodes were fabricated by PEC transformation via an ion-exchange process in the aqueous medium comprising two sulfur species (Na<sub>2</sub>S/Na<sub>2</sub>SO<sub>3</sub>) or (Na<sub>2</sub>S/Na<sub>2</sub>SO<sub>4</sub>). The PEC transformation process was carried out in a classical PEC cell, under visible-light excitation (100 mW/cm<sup>2</sup>) and cyclic voltammetry scans at 50 mV/s. BiVO<sub>4</sub> was electrodeposited over FTO glass (Hartford glass, 15 Ω/cm<sup>2</sup>) and was employed as the working electrode, while Pt and Ag/AgCl were the counter and reference electrodes. During this PEC treatment, the Bi<sub>2</sub>S<sub>3</sub> particles decorated over the BiVO<sub>4</sub> electrode surface. Lastly, the PEC ion-exchange process transformed yellow-colored BiVO<sub>4</sub> into black-colored BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> heterostructured photoanodes. The synthetic method employed for the PEC transformation of BiVO<sub>4</sub> into BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> heterostructured photoanodes is presented in Scheme S1.

### 2.4. Synthesis of BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>/BiPS<sub>4</sub> hybrid nanostructures

BiPS<sub>4</sub> was decorated over the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> heterostructured photoanodes via a subsequent PEC deposition process under visible-light excitation in a Na<sub>2</sub>S/PBS electrolyte solution. After the ion-exchange process in the aqueous solution, both BiPS<sub>4</sub> and Bi<sub>2</sub>S<sub>3</sub> particles decorated the surface of BiVO<sub>4</sub>. The proposed reaction mechanisms are detailed below:

(1. In solution)

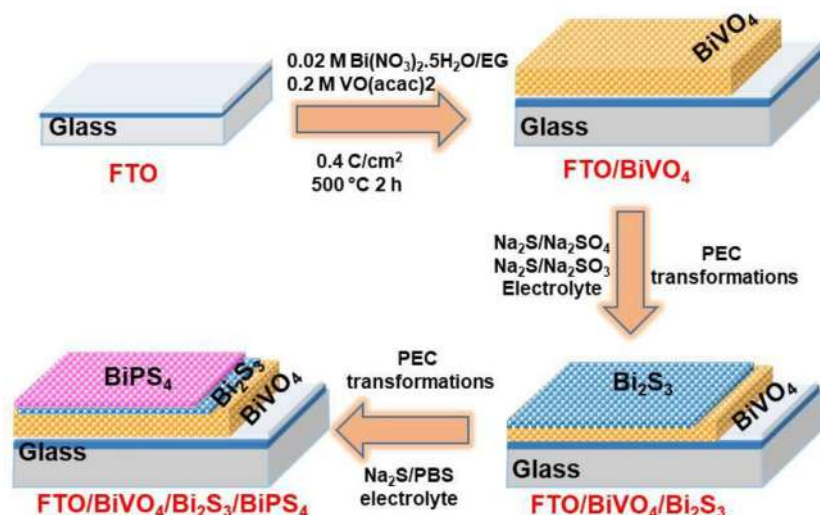


(2. Solid-liquid interface)



### 2.5. Characterization

X-ray Diffraction (XRD) spectra were executed with Rigaku Miniflex-600 equipment. Surface analysis was executed using XPS (Specs SAGE 150), Where a morphological examination of



**Scheme 1. Synthesis of Triple-layer heterostructured nanostructures.** Schematic illustration of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3/\text{BiPS}_4$  heterostructured film on an FTO deposited via a developed PEC ion-exchange process.

the electrodes was carried out by Field Emission SEM with a JEOL JSM-7000F system.

## 2.6. Photoelectrochemical measurements

PEC features of the obtained films were examined by assessing their photocurrent density through a water-splitting reaction. The photocurrent response was evaluated through a potentiostat (Autolab, PGSTAT30). PEC examination of the electrodes was carried out through CV and LSV under-stimulated light conditions ( $100 \text{ mW/cm}^2$ ) in  $0.1 \text{ M Na}_2\text{S/PBS}$  (pH 10). Also, electrochemical impedance spectroscopy (EIS) ( $100 \text{ kHz}$ – $100 \text{ mHz}$ ) study was obtained through an electrochemical system in  $0.1 \text{ M PBS}$  solution at  $0.65 \text{ V}_{\text{RHE}}$  under illumination situations.

## 3. Results and discussion

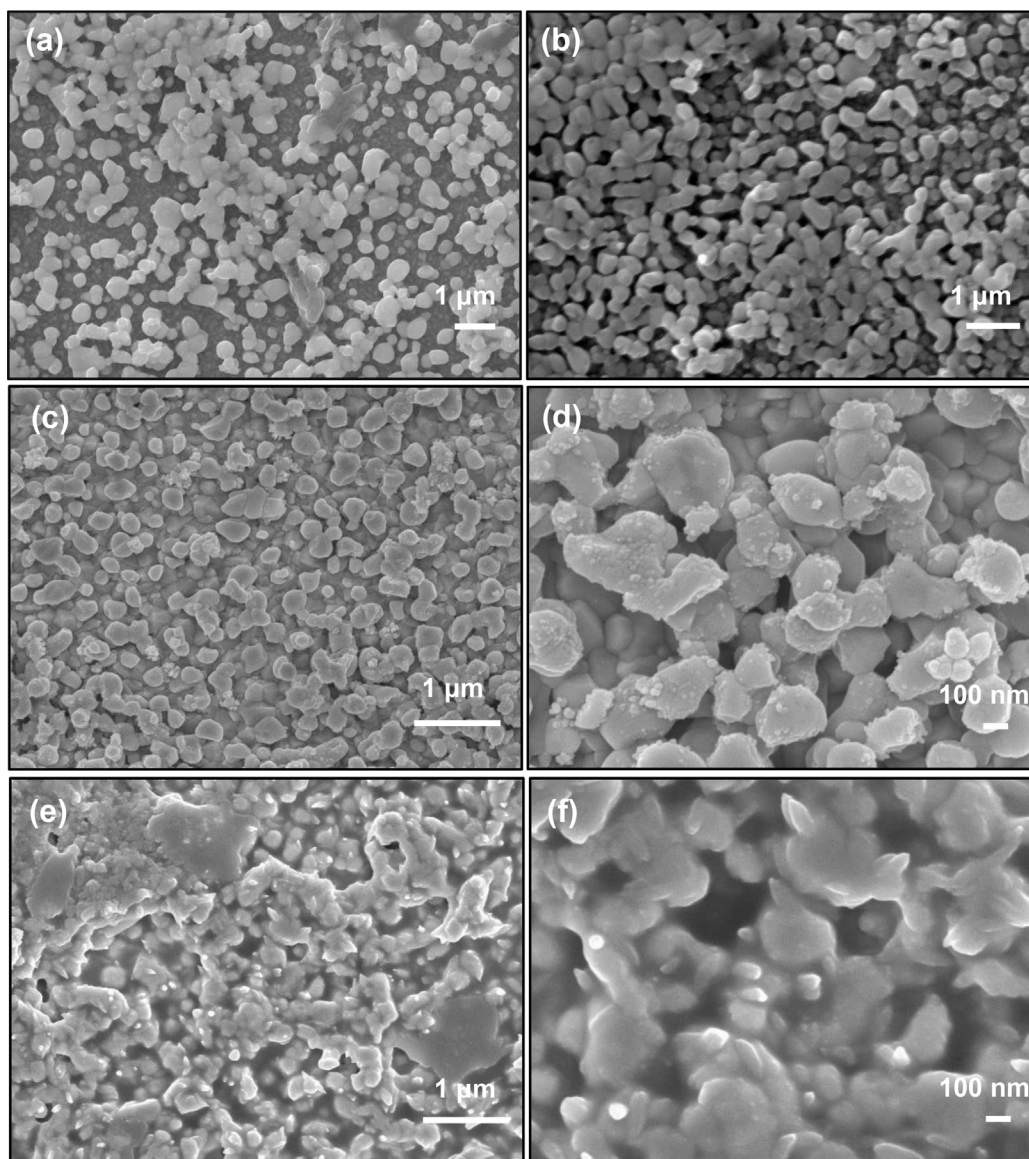
### 3.1. Materials characterization

Bismuth thiophosphate ( $\text{BiPS}_4$ ) was decorated over  $\text{BiVO}_4$  via ion-exchange reactions in an electrolyte comprising  $0.1 \text{ M PBS/Na}_2\text{S}$  solution. Particularly, Scheme S1 shows the sequential steps of the synthetic method for fabricating the  $\text{BiVO}_4/\text{Bi}_2\text{S}_3/\text{BiPS}_4$  photoanodes. Notably, the adsorbed water over the electrode surface generates  $\text{OH}^-$  ions during the CV scans. Also,  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$  and bare  $\text{BiVO}_4$  photoanodes were obtained using an electrodeposition process on FTO, as described in the experimental Methods.

The morphological characterization during the various stages of the synthetic process of the  $\text{BiVO}_4/\text{Bi}_2\text{S}_3\text{-BiPS}_4$  composite photoanodes was carried out through FE-SEM. The optimal  $\text{BiVO}_4$  photoanodes were initially transformed with  $\text{Bi}_2\text{S}_3$  catalysts by ion-exchange reactions in  $0.1 \text{ M Na}_2\text{SO}_3$ . For comparison, Fig. 1a, b shows the FE-SEM photographs of  $\text{BiVO}_4$  films; the anodes keep the smoother-edged dendrites with a large surface area. Fig. 1c displays the FE-SEM micrographs of the  $\text{Bi}_2\text{S}_3$  nanoparticles through ion exchange reactions (10 cycles) on the  $\text{BiVO}_4$  ( $\text{BiVO}_4/\text{Bi}_2\text{S}_3\text{-1}$ ) photoanodes prepared in  $0.1 \text{ M Na}_2\text{SO}_4 + \text{Na}_2\text{S}$  electrolytes. The FE-SEM images of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$  electrodes display direct evidence of  $\text{Bi}_2\text{S}_3$  loading was homogenous and very thin on the  $\text{BiVO}_4$  surface (Fig. 1d). Fabricated  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$  photoanodes were then transformed into  $\text{BiVO}_4/\text{Bi}_2\text{S}_3\text{-BiPS}_4$  by PEC treatment (via 10 CV scans)

in  $0.1 \text{ M PBS}$  (Fig. 1e). Finally, as seen in Fig. 1f, the FESEM photographs of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3\text{-BiPS}_4$  obviously disclose that the  $\text{BiVO}_4$  surface became homogenously smoother after incorporating phosphate.

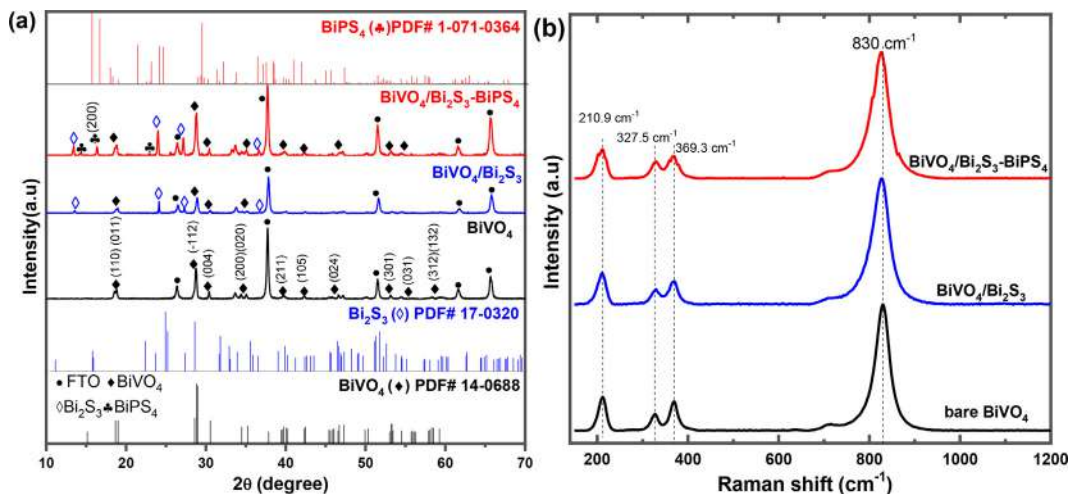
Fig. 2a illustrates the X-ray diffraction patterns for the bare  $\text{BiVO}_4$  before and after incorporating  $\text{Bi}_2\text{S}_3$  and  $\text{BiPS}_4$  ( $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ ,  $\text{BiVO}_4/\text{Bi}_2\text{S}_3/\text{BiPS}_4$ ) by the PEC ion-exchange process. There was substantial variation in the diffraction patterns of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$  electrodes, validating the development of new phases in the modified  $\text{BiVO}_4$  electrodes. For the bare  $\text{BiVO}_4$  (black), sharp peaks assigned to the  $\text{BiVO}_4$  electrode agreed with the JCPDS # 00–014–0688 and were denoted by the symbol ( $\blacklozenge$ ) [38–39]. After the introduction of  $\text{Bi}_2\text{S}_3$ , Fig. 2a (blue curve), the observed peaks agree with the  $\text{Bi}_2\text{S}_3$  phase, corresponding to the JCPDS pattern (00–017–0320) of bare  $\text{Bi}_2\text{S}_3$  [40–41], which were labeled as ( $\Delta$ ). Finally, the diffraction pattern of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3\text{-BiPS}_4$  hybrid nanostructures (red curve) validates the decoration of  $\text{BiPS}_4$  over  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ , corresponding to the JCPDS file No. (1–071–0364), which were labeled as ( $\heartsuit$ ) [42]. The diffraction patterns presented in Fig. 2a clearly display the presence of both  $\text{Bi}_2\text{S}_3$  and  $\text{BiPS}_4$  over the surface of  $\text{BiVO}_4$ . Further, Raman Spectroscopy was executed to assess the local structure of fabricated heterostructured photoanodes (Fig. 2b). The vibrational bands at  $\sim 210.9$ ,  $327$ ,  $369$ , and  $830 \text{ cm}^{-1}$  characteristics for  $\text{BiVO}_4$  were observed for all the fabricated photoanodes consistent with the earlier works [43–45]. Also, the characteristic vibrational bands situated at  $830 \text{ cm}^{-1}$  correspond to the asymmetric and symmetric V–O stretching vibration mode, whereas the other Raman peaks at  $210 \text{ cm}^{-1}$ ,  $326 \text{ cm}^{-1}$ , and  $371 \text{ cm}^{-1}$  were allocated to the external vibration mode and asymmetric–symmetric bending vibrations of vanadates. However, in the decorated samples, the observed blue shifts in the major Raman bands were related to symmetric V–O stretching, reflecting asymmetry in the crystalline structure and evidently witnessed in  $\text{BiVO}_4/\text{Bi}_2\text{S}_3\text{-BiPS}_4$  photoanodes. This electrode reveals the presence of the most intense Raman band owed to symmetric V–O stretching ca.  $830 \text{ cm}^{-1}$ , indicating a higher degree of disorder for average vanadate species [46]. On the other hand, the thin homogenous  $\text{Bi}_2\text{S}_3\text{-BiPS}_4$  can explain the absence of Raman modes that are specific to the  $\text{Bi}_2\text{S}_3\text{-BiPS}_4$  coating. Hence, we can conclude that the  $\text{Bi}_2\text{S}_3\text{-BiPS}_4$  hybrid heterostructures were successfully prepared through the electrochemical ion exchange treatment process with intimate modification to the  $\text{BiVO}_4$  surface.



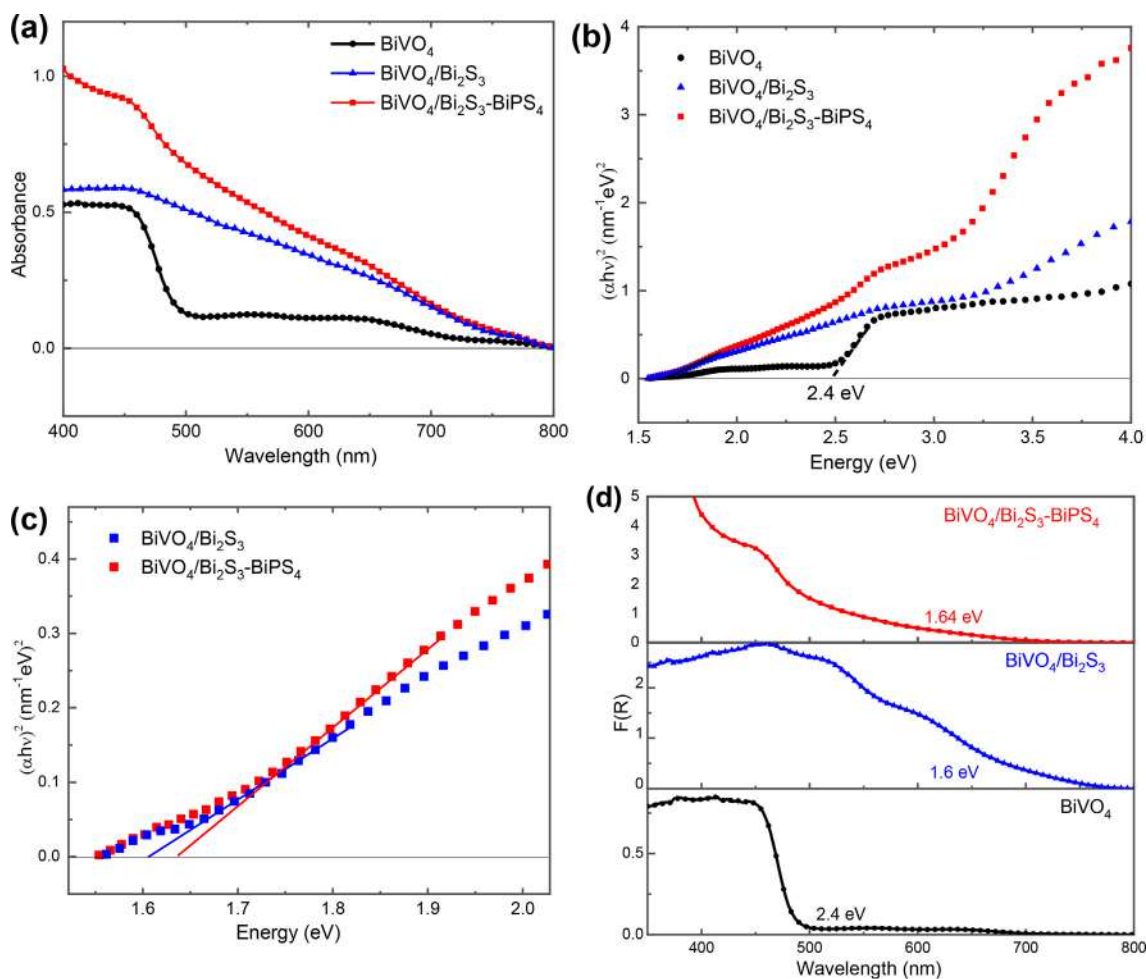
**Fig. 1.** FESEM micrographs of  $\text{BiVO}_4$  photoanodes. FE-SEM photographs of the (a,b)  $\text{BiVO}_4$  photoanodes obtained through electrochemical deposition process at different magnifications; (c,d)  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$  photoanodes obtained after 10 cycles in  $\text{Na}_2\text{SO}_3$  electrolyte, and (e, f)  $\text{BiVO}_4/\text{BiPS}_4\text{-Bi}_2\text{S}_3$  photoanodes obtained after 10 cycles under the treatment of  $\text{Na}_2\text{S}$  and PBS solution.

Fig. 3 illustrates UV–visible spectroscopy results of bare- $\text{BiVO}_4$  and  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ , and  $\text{BiVO}_4/\text{Bi}_2\text{S}_3\text{-BiPS}_4$  samples acquired after 10 cycles under the treatment of  $\text{Na}_2\text{S}$  and PBS solution. Fig. 3a displays the absorption spectra of the three electrodes, while Fig. 3b shows their Tauc plots, where the bandgap for  $\text{BiVO}_4$  was assessed to be 2.4 eV. It has to be mentioned here that the absorption spectra of both  $\text{Bi}_2\text{S}_3$  and  $\text{Bi}_2\text{S}_3\text{-BiPS}_4$  decorated electrodes showed long Urbach tails at the absorption onsets rendering it difficult to accurately estimate the bandgap edge from their Tauc plots. Fig. 3c shows a magnified image for both samples at the absorption onset region where the bandgap edge was assessed to be about 1.6 eV and 1.64 eV respectively for the  $\text{Bi}_2\text{S}_3$  and  $\text{Bi}_2\text{S}_3\text{-BiPS}_4$  decorated electrodes. Fig. 3d shows the correlation between  $F(R)$  and wavelength for fabricated electrodes. As shown in Fig. 3a, the light absorption capability of  $\text{BiVO}_4$  is considerably improved after  $\text{Bi}_2\text{S}_3$  was transformed. The combined deposition of  $\text{Bi}_2\text{S}_3$  and  $\text{BiPS}_4$  nanoparticles provide an additional synergetic enhancement of the optical density of the bare  $\text{BiVO}_4$  electrodes in the visible-light region. More importantly, it illustrates that

the decoration of the  $\text{BiPS}_4$  and  $\text{Bi}_2\text{S}_3$  particles over  $\text{BiVO}_4$  can lead to the absorption of a higher fraction of the visible portion of the solar spectrum. In particular, when  $\text{Bi}_2\text{S}_3$  was deposited, the extended absorption range is directly related to its narrow bandgap at  $\sim 1.6$  eV, where the visible spectrum between the absorption edge of  $\text{BiVO}_4$  at  $\sim 2.4$  eV and that of  $\text{Bi}_2\text{S}_3$  at  $\sim 1.6$  eV could now be absorbed. The large absorption coefficient of  $\text{Bi}_2\text{S}_3$  is another possible reason behind the improved light harvesting. Notably, recent reports have proved that the  $\text{Bi}_2\text{S}_3$  nanostructured particles have strong absorption over  $\text{BiVO}_4$  [47], as demonstrated by the improved optical absorbance at  $\lambda > 500$  nm in Fig. 3a. When  $\text{BiPS}_4$ , with its bandgap of  $\sim 1.64$  eV, was additionally introduced, the absorption onset in the absorption spectrum was clearly not affected due to the almost similar bandgap of  $\text{Bi}_2\text{S}_3$  previously deposited. However, the optical density was remarkably enhanced (almost doubled over the whole absorption region). Therefore, the absorption spectrum of the  $\text{BiVO}_4/\text{Bi}_2\text{S}_3\text{-BiPS}_4$  photoanodes was both stretched over a wider region and enhanced by a factor of  $\sim 2$ .



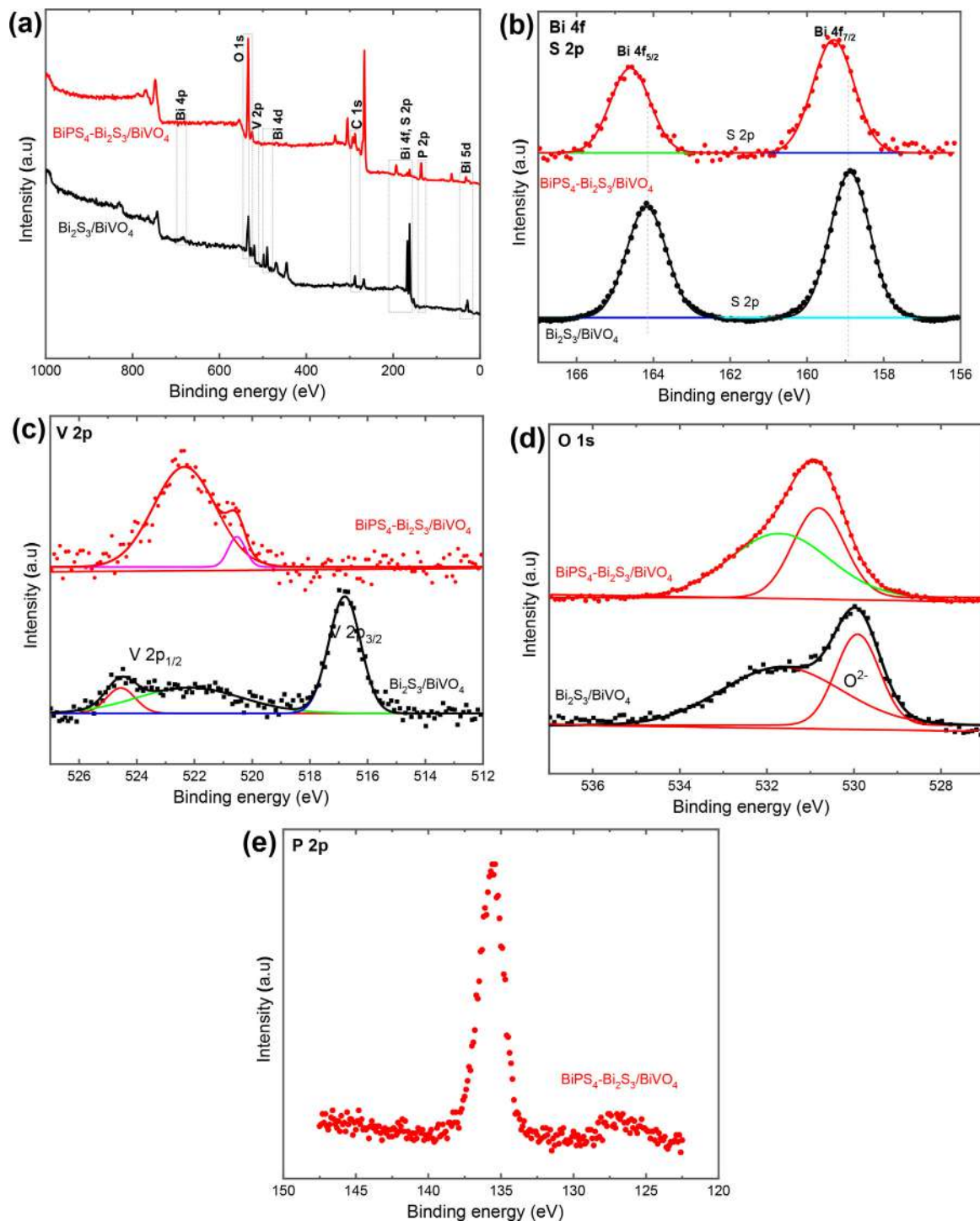
**Fig. 2.** Diffraction pattern of photoanodes. (a) Normalized X-ray diffraction pattern for BiVO<sub>4</sub>, BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> heterostructures, and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> hybrid heterostructures prepared over FTO substrate through electrochemical ion exchange reactions. (b) Raman spectra of fabricated BiVO<sub>4</sub>, BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>, and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub>, obtained in the region of 150–1200 cm<sup>-1</sup>.



**Fig. 3.** Optical features of photoanodes. (a) UV-vis absorption spectra of BiVO<sub>4</sub>, BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>, and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> hybrid heterostructures prepared by means of electrochemical ion-exchange reactions, (b) the corresponding correlation between  $(\alpha h\nu)^2$  and  $E$  (eV) for bare BiVO<sub>4</sub> and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>, and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> and their enlarged view (c) of electrodes, (d) the corresponding Kubelka-Munk function  $F(R)$  for bare BiVO<sub>4</sub> and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>, and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> with bandgaps of 2.4, 1.6, and 1.64 eV, respectively.

Moreover, the XPS study provides further insights into the interface between BiVO<sub>4</sub> and BiPS<sub>4</sub>-Bi<sub>2</sub>S<sub>3</sub>. The obtained spectra

are reported in Fig. 4. The full XPS survey spectra of Fig. 4a indicate that BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> contains Bi, V, O, P, and S. Also, the XPS



**Fig. 4.** Surface characterization of photoanodes. Survey XPS spectra of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3\text{-BiPS}_4$  and  $\text{Bi}_2\text{S}_3/\text{BiVO}_4$  electrodes (a), core level XPS spectra of Bi 4f (b), V 2p (c), O 1s (d), P 2p (e).

spectrum of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$  photoanodes before the PEC ion-exchange process in 0.1 PBS is included for comparison. The peaks observed at 158.7 and 163.7 eV are splitting peaks of  $\text{Bi}4f_{7/2}$  and  $\text{Bi}4f_{5/2}$ , individually, which are features of  $\text{Bi}^{3+}$  (Fig. 4b) [35]. However, the smaller signal at approximately 161.3 eV is attributed to the spin state of S  $2p_{1/2}$ . Likewise, the obtained peaks positioned at 522.0 ( $\text{V}2p_{1/2}$ ) and 516.8 eV ( $\text{V}2p_{3/2}$ ) are the splitting peaks of V2p, corresponding to the surface  $\text{V}^{5+}$  species (Fig. 4c). Furthermore, the acquired peaks at 529.9 and 531.9 eV in the O1s spectrum (Fig. 4d) correspond to O bonded inside an oxide crystal ( $\text{O}^{2-}$ ) in

bare- $\text{BiVO}_4$  electrodes and to that of adsorbed  $\text{OH}^-$  groups over the electrode surface, correspondingly [48]. More importantly, the high-resolution spectra of Bi (Fig. 4b), V (Fig. 4c), and O (Fig. 4d) of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3\text{-BiPS}_4$  are dissimilar compared to those for  $\text{BiVO}_4\text{-Bi}_2\text{S}_3$ , with a substantial shift to higher binding energy (BE) in Bi 4f, V2p, and O1s peaks in the heterostructured photoanodes. This evidences an effective efficient monolithic interface between  $\text{BiVO}_4$  and  $\text{BiPS}_4\text{-Bi}_2\text{S}_3$ . The higher BE shift is owing to the introduction of sulfur (which has a higher electron affinity) over the  $\text{VO}_4^{3-}$  tetrahedral sites of  $\text{BiVO}_4$  [49], consistent with the

Raman analysis above (Fig. 2b) [50]. The HR-XPS spectrum in Fig. 4e displays a P 2p signal at 135.7 eV, assigned to P–O and Bi–P–S combinations. These results indicate that the Bi<sub>2</sub>S<sub>3</sub> and BiPS<sub>4</sub> nanoparticles were effectively transformed during the PEC transformation process in 0.1 M PBS/Na<sub>2</sub>S (pH 10).

### 3.2. Photoelectrochemical features of photoanodes

A classical 3-electrode assembly was employed for PEC measurements under AM 1.5G simulated sunlight. The highly active BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> heterostructured electrodes from a PEC transformation of BiVO<sub>4</sub> in different electrolytes comprising 0.1 M Na<sub>2</sub>SO<sub>3</sub>/Na<sub>2</sub>SO<sub>4</sub> and 0.1 M Na<sub>2</sub>S solutions, carried out by varying the number of deposition cycles under light excitations at 100 mW/cm<sup>2</sup>. Fig. 5a illustrates the linear sweep voltammogram (LSV) plots of the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-1 (prepared in 0.1 M Na<sub>2</sub>SO<sub>4</sub> + 0.1 M Na<sub>2</sub>S electrolytes) photoanodes obtained by different photo deposition cycles in 0.1 M Na<sub>2</sub>S/Na<sub>2</sub>SO<sub>4</sub> solutions; cycle-10 shows the highest photocurrent, which was selected for fabrication of heterostructured photoanodes. Similarly, the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-2 electrodes were obtained in the 0.1 M Na<sub>2</sub>S/Na<sub>2</sub>SO<sub>3</sub> solutions, and the results are presented in Fig. 5b. The optimized number of electrochemical cycles for the maximal photocurrent density was 10 cycles. Under light excitation, the enhancement in photocurrent is believed to be due to the PEC transformation, the surface of BiVO<sub>4</sub> initiated the transformation into Bi<sub>2</sub>S<sub>3</sub> by the diffusion of V-ions into the Na<sub>2</sub>S/Na<sub>2</sub>SO<sub>3</sub> solution, and the diffusion of Sulfur into the BiVO<sub>4</sub> lattice [51]. Afterward, the observed photocurrent decrease in the optimal Bi<sub>2</sub>S<sub>3</sub> films developed over the BiVO<sub>4</sub> surface, which contains the irregular block to the V-leaching process. Further, BiPS<sub>4</sub> was decorated over BiVO<sub>4</sub> via ion-exchange reactions in an electrolyte comprising 0.1 M PBS and 0.1 M Na<sub>2</sub>S solution (Fig. 5c). The PEC conditions for obtaining the best Na<sub>2</sub>S/PBS mixing of the nanoparticles were improved by changing electrochemical cycles. Likewise, the addition of BiPS<sub>4</sub> on the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> films was activated under ion-exchange reactions under light excitations, and the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> (10 cycles) showed the highest PEC performance (Fig. 5c). In addition to the enhanced light absorption, we believe that the integration of BiVO<sub>4</sub> and Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> created a triple heterostructured junction, resulting in more efficient photoinduced carrier separation and declined charge recombination [52–53].

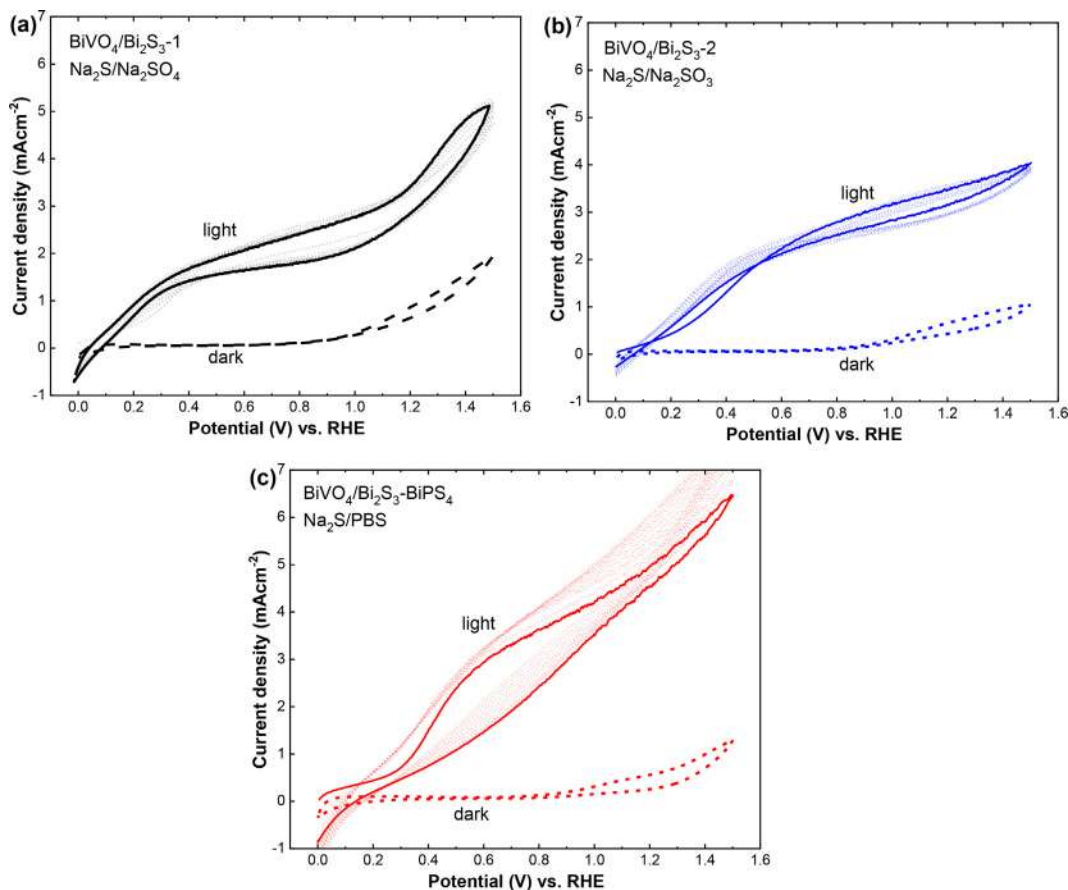
The distinctive photocurrent–potential (J–V) schemes for the PEC water splitting reaction in Fig. 6a show that the optimal BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> photoanodes presented superior photocurrents to BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-1 and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-2. The high onset potential (0.85 V<sub>RHE</sub>) and lower photocurrent response of the bare-BiVO<sub>4</sub> (~0.7 mA/cm<sup>2</sup> at 1.23 V<sub>RHE</sub>) are apparently credited to the poor hole transfer kinetics at the interface between the bare BiVO<sub>4</sub> and water [54]. The photocurrent density upsurged from 1.97 to 3.7 mA/cm<sup>2</sup> after introducing the BiPS<sub>4</sub> layer owing to enriched light absorption and, successively, enhanced photoinduced carrier generation. Further, the photocurrent density considerably upsurged with Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> decoration and attained ~3.7 mA/cm<sup>2</sup> at 1.23 V<sub>RHE</sub> in 0.1 M PBS/Na<sub>2</sub>S (pH 10), which is nearly a 5-fold enrichment compared to bare BiVO<sub>4</sub>. The photocurrent onset potentials (V<sub>on</sub>) for the fabricated heterostructured electrodes were estimated from the quasi-steady-state J–V schemes acquired at a photocurrent density of 1 mA/cm<sup>2</sup>, and the estimated values were 0.25 V<sub>RHE</sub> for BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> compared 0.85 V<sub>RHE</sub> for BiVO<sub>4</sub> (Fig. 6a). Compared to the other samples, a superior photocurrent density was achieved in the lower potential area (0.6 V<sub>RHE</sub>). Based on this fact, it is obvious that a transformation toward BiPS<sub>4</sub> and Bi<sub>2</sub>S<sub>3</sub> films is mandatory to justify the demand for higher efficiency as well as durability. For estimating the ABPE efficiencies of the heterostructured electrodes, the raw data were obtained from

Fig. 6a, and are presented in Fig. 6b. The maximal photoconversion efficiency of the fabricated BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> heterostructured photoanode is (1.40 % at 0.60 V<sub>RHE</sub>), which is substantially nearly 2-fold enhancement compared to the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-1 photoanode (0.79 % at 0.6 V<sub>RHE</sub>) and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-1 photoanode (0.81 % at 0.6 V<sub>RHE</sub>). Fig. 6c displays the chopped LSVs of the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub>, BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-1, BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-2 at 5 mV/s under pulsed light irradiation in 0.1 M Na<sub>2</sub>S/PBS solution (pH ~ 10). Fig. 6e showed the linear relationship between ΔJ = (J<sub>a</sub>–J<sub>c</sub>) at + 1.23 V<sub>RHE</sub> and the sweep rates; from the slope of the linear fit, the C<sub>dl</sub> was estimated and shown in Table 1, which is openly related to the electrochemically active surface area of the materials [55]. The BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> electrodes (52.65 μF/cm<sup>2</sup>) have a higher C<sub>dl</sub>, which is a 3-fold enhancement than the bare BiVO<sub>4</sub> (19.6 μF/cm<sup>2</sup>), which evidences that the integration of Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> over the surface of the BiVO<sub>4</sub> advances the active surface area and develops more active sites for water oxidation. Comparatively, a superior photocurrent density was acquired BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> in the lower potential part (0.6 V<sub>RHE</sub>) than the other electrodes (Fig. 6f), which is remarkable for developing tandem photoanode-photocathode arrangements.

The dynamics of charge transfer features of the heterostructured electrodes were explored by EIS under visible-light excitation. The Nyquist curve of the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> electrodes evaluated under visible-light excitation at 0.65 V<sub>RHE</sub> and the matching equivalent circuit are exhibited in Fig. 7a. Notably, these results validate that the diameter of the arc on the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> heterostructured photoanodes was lower than that of BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> electrodes. The smaller arc radius of the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> electrodes suggests faster interfacial charge transfer than bare BiVO<sub>4</sub>. Notably, the fabricated BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> heterostructured electrode shows the smallest R<sub>ct</sub> value (Table S1). The constant capacitance and reduction in the charge transfer resistance (R<sub>CT</sub>) at similar voltages reveal that BiPS<sub>4</sub> functioned as an efficient film, boosting the charge transfer kinetics and reducing surface recombination. These results validate that the PEC surface transformation of BiVO<sub>4</sub> to Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> can reduce the R<sub>CT</sub> considerably, due to the development of a triple BiPS<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>/BiVO<sub>4</sub> heterostructured interface system.

Fig. 7b demonstrates the corresponding Bode phase curves of the BiVO<sub>4</sub>, BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>, and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub>, which explains the kinetics of the electrochemical reaction on the electrode surface [56]. The higher frequency of the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> submits the quicker water oxidation kinetics via the rapid hole transfer process. Moreover, the hole relaxation lifetime (τ<sub>p</sub>) was estimated from the Bode phase plot and provided in Table 1. The smallest τ<sub>p</sub> of the BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> photoanode evidence the rapid hole-consumed water oxidation process. Likewise, from τ<sub>p</sub>, the hole diffusion length of the BiVO<sub>4</sub> and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> photoanodes is also derived and given in Table 1, supporting the greater hole consumed water oxidation kinetics in BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> than BiVO<sub>4</sub> [57]. All the acquired results noticeably show that the transformation of both Bi<sub>2</sub>S<sub>3</sub> and BiPS<sub>4</sub> over BiVO<sub>4</sub> electrodes influences the PEC features deeply.

The long-term durability of the heterostructured electrode was inspected by chronoamperometric measurements. Fig. 8a displays the long-standing durability of BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> at 0.8 V<sub>RHE</sub> built on the J–t plots obtained in 0.1 M PBS/Na<sub>2</sub>S over 10 h, at a photocurrent response of ~3.4 mA/cm<sup>2</sup>; there was no apparent decay, demonstrating the remarkable durability with the integration of BiPS<sub>4</sub> under continuous illumination. Particularly, the obtained photocurrent parameters agreed with the results acquired from the equivalent LSV curves. The heterostructured BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> heterostructured photoanodes, which exhibit the synergetic effect of both Bi<sub>2</sub>S<sub>3</sub> and BiPS<sub>4</sub>, exhibited a faster decline of photocurrent at first and could withstand a substantial photocurrent density after 7 h of about 2.84 mA/cm<sup>2</sup>, which is ~ 84 % of its initial



**Fig. 5.** Effect of electrolytes on the PEC activity of  $\text{BiVO}_4$  electrodes. (a) CV characteristics of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ -1 photoanodes prepared in 0.1 M  $\text{Na}_2\text{SO}_4$  + 0.1 M  $\text{Na}_2\text{S}$  electrolytes (pH 10) under visible light excitations conditions ( $100 \text{ mW}/\text{cm}^2$ ) for continuous 10 cycles, (b) CV characteristics of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ -2 photoanodes prepared using 0.1 M  $\text{Na}_2\text{S}/\text{Na}_2\text{SO}_3$  electrolyte (pH 10) under irradiation of and (c) CV characteristics of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ - $\text{BiPS}_4$  photoanodes prepared through 0.1 M  $\text{Na}_2\text{S}$  + 0.1 M PBS electrolyte (pH 10) under light excitations. Dark (dashed line), Light (dotted line), after continuous cycles (solid line).

value. However, the J-t profile of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ - $\text{BiPS}_4$  ultimately revealed better durability compared to that of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$  after 10 h of testing, as can be revealed from the enlarged J-t plot in Fig. 8b. Moreover, the  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ - $\text{BiPS}_4$  heterostructured photoanodes present less spikes on the photocurrent than the  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$  photoanodes. These results validate the suppressed charge recombination sites for the photoanodes with the introduction of  $\text{BiPS}_4$ . In addition, the photoelectrochemically transformed  $\text{BiPS}_4$  layer over the  $\text{BiVO}_4$  surface will generate additional photoinduced charge carriers with an additional type II  $\text{Bi}_2\text{S}_3/\text{BiPS}_4$  heterojunction for efficient charge separation. Our results, therefore, validate the function of the transformed electrodes in suppressing the charge recombination and boosting the PEC features at the same time [58]. Table 2 relates the photoanodes fabricated in the present work with those stated earlier in the literature. The results designate that our  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ - $\text{BiPS}_4$  heterostructured electrodes are durable and efficient for oxidizing water into  $\text{O}_2$ .

Photoelectrochemical production of OER via photoelectrocatalysis at  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$  and  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ - $\text{BiPS}_4$  films were observed through an Oxsense instrument. Figure S1 shows the evidence of  $\text{O}_2$  evolution concentration with respect to the time for triple-layered  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ - $\text{BiPS}_4$  electrodes at  $0.6 V_{\text{RHE}}$  and with continuous illumination. As observed in the figures, after  $\text{BiPS}_4$  addition, visible-light photons ( $\lambda > 420 \text{ nm}$ ) insisted dioxygen evolution over the electrode surface of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ - $\text{BiPS}_4$ , and their matching photocurrent analysis is shown in Fig. S1b. With the aid of potential ( $0.6 V_{\text{RHE}}$ ), dioxygen generation was detected, and their acquired concentrations were expected to increase linearly with time inter-

vals. On the other hand, the obtained  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ - $\text{BiPS}_4$  electrodes evidenced considerable durability towards continuous illumination conditions (Fig. S1b). All the acquired results evidence that the synergetic effect of  $\text{BiPS}_4$  and  $\text{Bi}_2\text{S}_3$  transformation considerably boosted the PEC performances and durability of  $\text{BiVO}_4$  electrodes by improving the charge-carrier density and the surface oxidation kinetics.

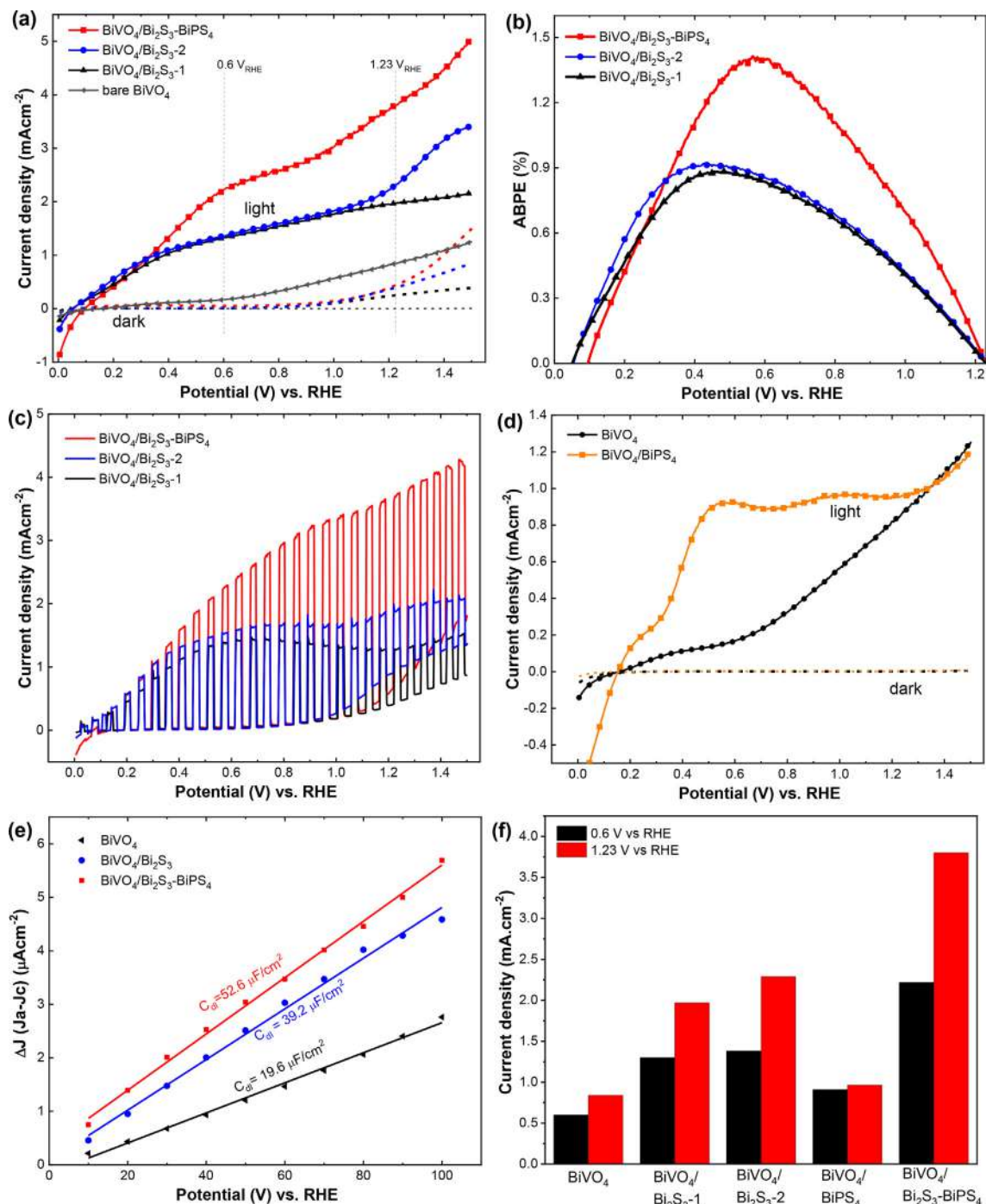
Generally, the PEC features of prepared electrode materials rely mainly on their energy band structures [70]. The energy level positions of the conduction and valence band can be predicted by the following equations [71]

$$E_{\text{CB}} = X - E_e - 0.5E_g \quad (1)$$

$$E_{\text{VB}} = E_{\text{CB}} + E_g \quad (2)$$

where X and  $E_g$  are the absolute electronegativity and the band-gap of the materials, correspondingly.  $E_e$  is the energy of a free electron on the hydrogen scale (4.5 eV). The X values for bare  $\text{BiVO}_4$ ,  $\text{Bi}_2\text{S}_3$ , and  $\text{BiPS}_4$  are 6.04 [72], 5.26 [73], and 5.31 eV, individually. The acquired bandgap parameters for  $\text{BiVO}_4$ ,  $\text{Bi}_2\text{S}_3$ , and  $\text{BiPS}_4$  have been assessed to be 2.4, 1.39, and 1.30 eV, correspondingly (Figure S2). Hence, the obtained ECB values of  $\text{BiVO}_4$ ,  $\text{Bi}_2\text{S}_3$ , and  $\text{BiPS}_4$  are assessed to be 0.34, 0.065, and 0.11 eV, individually. The  $E_{\text{VB}}$  values of  $\text{BiVO}_4$ ,  $\text{Bi}_2\text{S}_3$ , and  $\text{BiPS}_4$  can be assessed to be 2.74, 1.455, and 1.41 eV, respectively. These data determinations might assist to discuss the relative determinations of their energy bands (Figure S3). Notably, Fig. 9 illustrates the proposed mechanism of  $\text{BiVO}_4/\text{Bi}_2\text{S}_3$ - $\text{BiPS}_4$  heterostructured photoanode under





**Fig. 6.** (a) Polarization curves for BiVO<sub>4</sub>, BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-1 (0.1 M Na<sub>2</sub>SO<sub>4</sub>/0.1 M Na<sub>2</sub>S), BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-2 (0.1 M Na<sub>2</sub>SO<sub>3</sub>/0.1 M Na<sub>2</sub>S) and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> using a 0.1 M Na<sub>2</sub>S and 0.1 M PBS, (b) estimated ABPE efficiency, (c) corresponding chopped polarization curves in 0.1 M PBS, and (d) polarization curves in the dark (black) and under irradiation (orange) for BiVO<sub>4</sub>/BiPS<sub>4</sub> and (e) curves of ΔJ at 1.23 V<sub>RHE</sub> vs sweep rates; LSV curves of the obtained electrode (f) Variation in the photocurrent density for all fabricated materials in 0.1 M PBS and 0.1 M Na<sub>2</sub>S (pH 10) at 0.6 and 1.23 V<sub>RHE</sub>. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

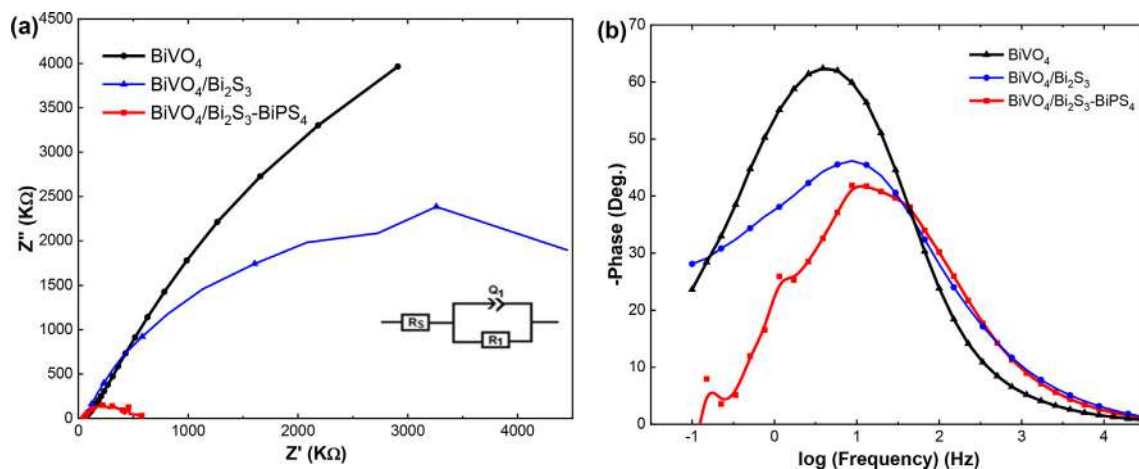
illumination conditions. Particularly, the photoinduced holes in the valence band of bare-BiVO<sub>4</sub> electrodes interact with the SO<sub>3</sub><sup>2-</sup>/S<sub>2</sub><sup>2-</sup> radicals in the presence of Na<sub>2</sub>S/Na<sub>2</sub>SO<sub>3</sub> sacrificial reagents and oxidize to develop S<sub>2</sub><sup>2-</sup> and S<sub>2</sub>O<sub>3</sub><sup>2-</sup>. Simultaneously, S<sub>2</sub><sup>2-</sup> ion effectively interacts with bare-BiVO<sub>4</sub> and produces Bi<sub>2</sub>S<sub>3</sub> at the surface of BiVO<sub>4</sub>. Firstly, the BiVO<sub>4</sub> film facilitates direct electron transfer, whereas Bi<sub>2</sub>S<sub>3</sub> assists as the front light absorber. As defined in the energy level band diagram of Fig. 9, the photogenerated electrons in the CB of Bi<sub>2</sub>S<sub>3</sub> transfer to the CB of BiVO<sub>4</sub>, and meanwhile the

photoinduced electrons are produced in the CB of BiVO<sub>4</sub> itself. Simultaneously, the photoinduced holes in the VB of BiVO<sub>4</sub> will effectively transport to the VB of Bi<sub>2</sub>S<sub>3</sub> owing to the built-in electric field developed by the type II heterojunction [65,74]. In a nutshell, electrons and holes will be produced in both materials, and the type II structure of the interface will drive electrons toward BiVO<sub>4</sub> and holes toward the water interface. The same can be mentioned for the additional BiPS<sub>4</sub> later in the triple structure. The three materials BiVO<sub>4</sub>, Bi<sub>2</sub>S<sub>3</sub>, and BiPS<sub>4</sub> make perfect valence band energy

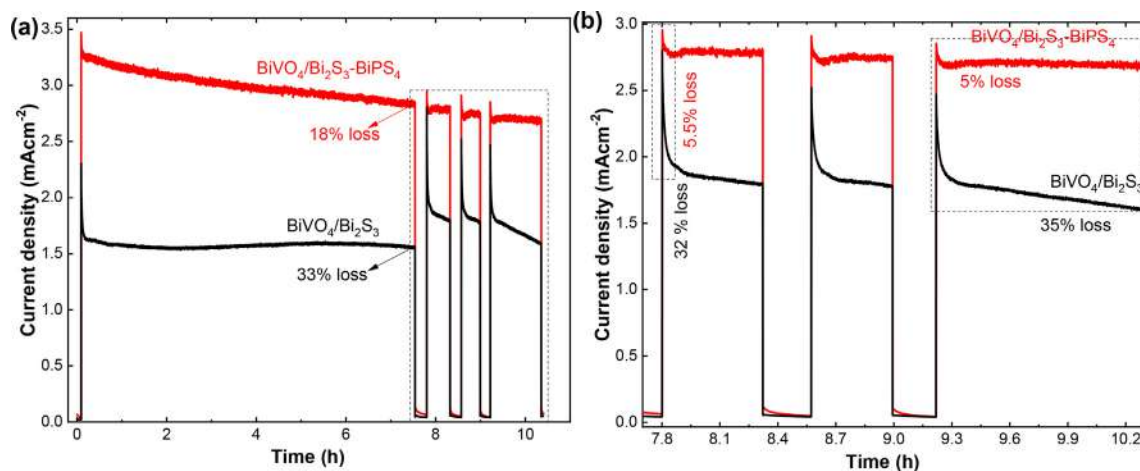
**Table 1**PEC values were acquired for the BiVO<sub>4</sub>, BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> electrodes.

Photoanode	J <sub>H<sub>2</sub>O</sub> (mA.cm <sup>-2</sup> ) @1.23 V <sub>RHE</sub>	ABPE (%)	R <sub>CT</sub> (Ω)	C <sub>dl</sub> (μF.cm <sup>-2</sup> )	Relaxation frequency (Hz)	τ <sub>p</sub> (ms)	L <sub>D</sub> (μm)
BiVO <sub>4</sub>	0.85	–	1.1 E <sup>12</sup>	19.6	4.624	34.43678	423.2
BiVO <sub>4</sub> /Bi <sub>2</sub> S <sub>3</sub>	1.97	0.89	6351	39.2	9.521	16.72468	294.9
BiVO <sub>4</sub> /Bi <sub>2</sub> S <sub>3</sub> -BiPS <sub>4</sub>	3.85	1.41	456.1	52.6	15.346	10.37636	232.3

Footnote: J<sub>H<sub>2</sub>O</sub> referring to the photocurrent density; ABPE - Applied bias photon-to-current efficiency; C<sub>dl</sub> - Double-layer capacitance; τ<sub>p</sub> -hole relaxation lifetime; L<sub>D</sub>- hole diffusion length;



**Fig. 7. Impedance spectra of photoanodes.** Comparative Nyquist plots (a) and its corresponding Bode phase plots (b) for BiVO<sub>4</sub> (black), BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> (blue) (tested in 0.1 M Na<sub>2</sub>SO<sub>3</sub> + 0.1 M Na<sub>2</sub>S), and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> (red) (tested in 0.1 M PBS + 0.1 M Na<sub>2</sub>S) (red) photoanodes at 1.23 V<sub>RHE</sub> with a frequency ranging between 100,000 to 0.05 Hz under visible-light excitations. Inset displays fitted equivalent circuits for the photoanodes. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)



**Fig. 8.** (a) J-t curve for long-term photodurability of BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> (tested in 0.1 M PBS + 0.1 M Na<sub>2</sub>S) (red) and BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> (tested in 0.1 M Na<sub>2</sub>SO<sub>3</sub> + Na<sub>2</sub>S) (black) at 0.8 V<sub>RHE</sub> for ~ 10 h under AM 1.5 G irradiation. (b) enlarged view of prepared electrodes. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

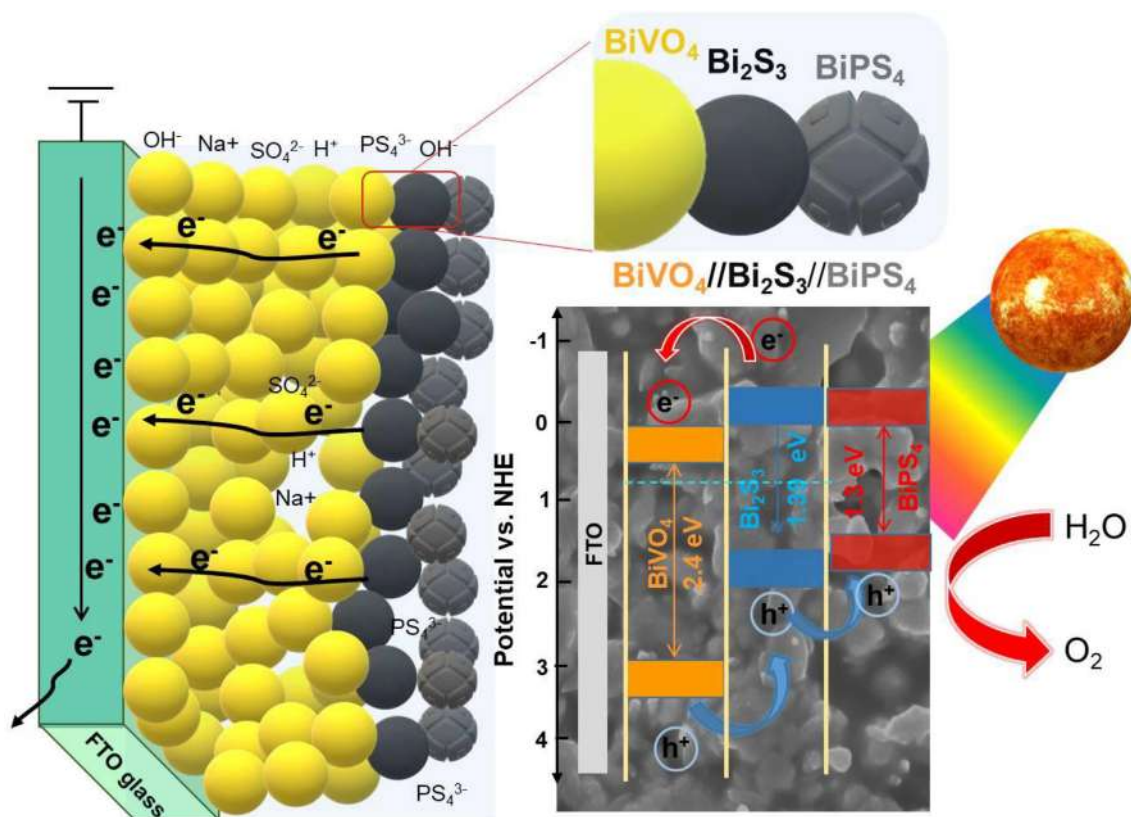
alignment for the efficient extraction of holes towards the water interface. Therefore, we have two types II heterojunctions and three absorbing materials. Hence, light absorption will be magnified and the charge transfer kinetics will be facilitated at the same time, explaining the large enhancement in the triple electrode. Hence, this triple-layer heterostructured BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> configuration can be an effective water oxidation process. These results provide basic knowledge and design of effective heterostructured photoanodes for solar fuel production.

#### 4. Conclusions

In summary, a noble BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> electrode was fabricated on a conducting substrate via the PEC ion exchange process. Initially, homogeneous BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub> particles are synthesized by the PEC transformation process in an aqueous Na<sub>2</sub>S/Na<sub>2</sub>SO<sub>3</sub> electrolyte and then subjected to the PEC -ion exchange process to produce BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> electrode in Na<sub>2</sub>S/PBS electrolyte. When the surface of BiVO<sub>4</sub> was transformed with the Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> layer, the resulting BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> heterostructured electrodes demon-

**Table 2**  
Various kinds of BiVO<sub>4</sub>-based electrode materials are loaded with different co-catalysts and their PEC properties for water-splitting reactions.

S.No	Electrode	Electrolyte (pH)	Co-catalyst (Method)	Light source	Current density (mA/cm <sup>2</sup> )	Ref.
1	BiVO <sub>4</sub> /FeVO <sub>4</sub>	0.2 M Na <sub>2</sub> SO <sub>4</sub> pH = 7	Electrospray technique	100 mW.cm <sup>-2</sup>	0.4 @ 1.23 V <sub>RHE</sub>	[59]
2	BiVO <sub>4</sub> /CoFe-NiOOH	0.5 M Na <sub>2</sub> SO <sub>4</sub> pH = 7	Lifting method/chemical process	100 mW/cm <sup>2</sup>	1.54 @ 1.23 V <sub>RHE</sub>	[60]
3	BiVO <sub>4</sub> /rGO/NiFe	0.5 M Na <sub>2</sub> SO <sub>4</sub> pH = ~6.9	Potentiostatic electrodeposition	100 mW.cm <sup>-2</sup>	1.30 @ 1.23 V <sub>RHE</sub>	[61]
4	BiVO <sub>4</sub> /Ni <sub>0.5</sub> Fe <sub>0.5</sub> -LDH	0.5 M Na <sub>2</sub> SO <sub>4</sub>	Electrodeposition process	100 mW.cm <sup>-2</sup>	1.21 @ 1.23 V <sub>RHE</sub>	[62]
5	CoPi/BiVO <sub>4</sub>	0.5 M Na <sub>2</sub> SO <sub>4</sub>	Photodeposition	100 mW.cm <sup>-2</sup>	1.1 @ 1.23 V <sub>RHE</sub>	[63]
6	CoFe-PB/BiVO <sub>4</sub>	0.1 M KPi	Wet processing method	100 mW.cm <sup>-2</sup>	1.0 @ 1.23 V <sub>RHE</sub>	[64]
7	Ag/Ni-Zr:BiVO <sub>4</sub>	0.1 M PBS pH 7.5	Electrochemical deposition process	100 mW.cm <sup>-2</sup>	3.14 @1.23 V <sub>RHE</sub>	[26]
8	NiFePB/Zr:BiVO <sub>4</sub>	0.1 M PBS pH 7.5	Electrodeposition process	100 mW.cm <sup>-2</sup>	3.23 @1.23 V <sub>RHE</sub>	[25]
9	BiVO <sub>4</sub> /Bi <sub>2</sub> S <sub>3</sub>	0.5 M Na <sub>2</sub> SO <sub>4</sub>	Photoassisted electrodeposition process	100 mW.cm <sup>-2</sup>	1.43 @1.23 V <sub>RHE</sub>	[35]
10	BiVO <sub>4</sub> /Bi <sub>2</sub> S <sub>3</sub>	0.35 M Na <sub>2</sub> SO <sub>3</sub> /0.25 M Na <sub>2</sub> S	PEC transformation	100 mW.cm <sup>-2</sup>	3.3 @1.23 V <sub>RHE</sub>	[65]
11	BiVO <sub>4</sub> /Bi <sub>2</sub> S <sub>3</sub> /FeOOH	0.1 M Na <sub>2</sub> SO <sub>4</sub>	Hydrothermal process	100 mW.cm <sup>-2</sup>	0.8 @0.4 V <sub>SCE</sub>	[66]
12	Mo:BiVO <sub>4</sub>	0.1 M Na <sub>2</sub> SO <sub>4</sub>	Pulsed laser deposition	100 mW.cm <sup>-2</sup>	2.1@1.23 V <sub>RHE</sub>	[67]
13	BiVO <sub>4</sub> /V-NiOOH/FeOOH	1 M KBi	Hydrothermal method	100 mW.cm <sup>-2</sup>	5.43 @ 1.23 V <sub>RHE</sub>	[68]
14	Activated EL-BiVO <sub>4</sub>	1 M KBi	In-situ potentiostatic photopolarisation	100 mW.cm <sup>-2</sup>	4.6 @1.23 V <sub>RHE</sub>	[69]
13	BiVO <sub>4</sub> /Bi <sub>2</sub> S <sub>3</sub>	0.1 M PBS/Na <sub>2</sub> S pH 10	Ion-exchange reactions	AM 1.5G/100 mW/cm <sup>2</sup>	1.97 @1.23 V <sub>RHE</sub>	This work
14	BiVO <sub>4</sub> /Bi <sub>2</sub> S <sub>3</sub> /BiPS <sub>4</sub>	0.1 M PBS/Na <sub>2</sub> S pH 10	Ion-exchange reactions	AM 1.5G/100 mW/cm <sup>2</sup>	3.85 @1.23 V <sub>RHE</sub>	This work



**Fig. 9.** Mechanistic illustration of BiVO<sub>4</sub>/Bi<sub>2</sub>S<sub>3</sub>-BiPS<sub>4</sub> heterostructured electrode and its energy level diagrams under irradiation for the electrodes and electrolyte interface.

strated superior photocurrent features of ~3.7 mA/cm<sup>2</sup> at 1.23 V<sub>RHE</sub>. The impedances and absorption spectra show that the BiPS<sub>4</sub> layer assists as a surface protective film and boosts the PEC

features by upsurging the photon absorption and carrier separation of the BiVO<sub>4</sub> film. The exceptional PEC feature and durability and the rapid, convenient, and well-regulated commercial fabrication

of the BiPS<sub>4</sub> photoanodes make it a favorable substitute for conventional metal oxide-protected water-splitting photoanodes.

### CRedit authorship contribution statement

**Maged N. Shaddad:** Conceptualization, Data curation, Formal analysis, Investigation. **Prabhakarn Arunachalam:** Conceptualization, Data curation, Formal analysis, Investigation, Supervision, Writing – original draft, Writing – review & editing. **Mahmoud Hezam:** Investigation, Conceptualization, Writing – review & editing. **Norah M BinSaedan:** Formal analysis. **Sixto Gimenez:** Supervision. **Juan Bisquert:** Supervision. **Abdullah M. Al-Mayouf:** Supervision, Funding acquisition, Project administration.

### Data availability

Data will be made available on request.

### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

### Acknowledgments

This project was funded by the National Plan for Science, Technology and Innovation, King Abdulaziz City for Science and Technology, Kingdom of Saudi Arabia, Award Number (14-NAN2323-02).

### Appendix A. Supplementary material

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.jcat.2022.12.032>.

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